

Preparation and characterization of $\text{ZrO}_2\text{-Al}_2\text{O}_3$ particulate nanocomposites produced by plasma technique

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Received 2 June 2006, in revised form 14 September 2006

Abstract. Nanosized spherical $\text{ZrO}_2(\text{Y}_2\text{O}_3)\text{-Al}_2\text{O}_3$ particulate composites with average particle size in the range of 30–40 nm have been prepared by evaporation of coarse grained oxides in an inductively coupled plasma. The powders contain m-, t- ZrO_2 and θ -, δ - Al_2O_3 phases. The presence of Al_2O_3 reduces the crystallite size of ZrO_2 and increases the content of the t- ZrO_2 phase and prevents phase transition of ZrO_2 during additional calcinations. The surface characteristics of $\text{ZrO}_2\text{-Al}_2\text{O}_3$ particles testify that zirconia particles are covered by alumina.

Key words: nanosized zirconia-alumina, plasma technique, phase composition, particle, crystallite size.

1. INTRODUCTION

Zirconia-alumina composites find wide application as high-temperature materials in cutting tools, ball bearings and furnace materials due to their high melting point, mechanical properties, fracture toughness, corrosion and shock resistance [^{1,2}]. At present zirconia-alumina composites are commercially available, but materials produced from such composites show poor microstructural homogeneity, particularly with respect to the grain size and phase distribution [³]. A promising approach to the improvement of the microstructure and homogeneity of the composite material is application of homogeneous nanosized $\text{ZrO}_2\text{-Al}_2\text{O}_3$ powders and their fast sintering [^{4,5}]. To obtain the finest grain size and possibly high homogeneity of components, several methods of preparation of nanosized powders have been developed [^{6,7}]. The $\text{ZrO}_2\text{-Al}_2\text{O}_3$ powders have

been prepared by the hydrolysis of zirconium propoxide in a dispersion of α -alumina particles in ethanol [8] by simultaneous hydrolysis of zirconium and aluminium isopropoxides [9], by controlled hydrolysis of chlorides [10] or by a hydrothermal process [4]. Nanosized $\text{ZrO}_2\text{-Al}_2\text{O}_3$ particulate composites with uniform distribution of components have been obtained by these methods. However, the wet chemical methods of synthesis of $\text{ZrO}_2\text{-Al}_2\text{O}_3$ powders include many stages and the used raw materials are usually expensive. Besides this, the production rates are relatively low.

Another approach is connected with the synthesis of nanosized powders from the vapour phase by using laser evaporation [11] or the thermal plasma technique [7]. The laser evaporation technique provides preparation of nanosized powders with narrow particle size distribution, but production rate is low in comparison with the plasma technique.

The aim of the present work is to develop a single-step plasma technique for producing highly homogeneous zirconia-alumina particulate composites and to study their characteristics.

2. EXPERIMENTAL

The thermal plasma technique is based on the evaporation of coarse-grained commercially available powders in inductively coupled air plasma using the technological equipment described in [12]. The zirconia and alumina powders with a particle size in the range of 10–30 μm are mixed and injected into the plasma flame by the carrier gas. Evaporation of raw oxides is achieved by varying the power of the RF oscillator, the flow rate of the plasma forming gas, feeding rate of the powder and their injection velocity. The formation of particles from vapours and their growth are controlled by introducing cold air into the reaction chamber.

Chemical and phase composition of the prepared oxides are determined by conventional chemical and X-ray diffraction analysis. The specific surface area is determined by argon adsorption-desorption method and the average particle size is calculated from these data. The crystallite size of zirconia phases is determined by the X-ray line broadening method using the Scherrer equation. The particle size and surface morphology are studied by scanning electronic microscopy (SEM) and X-ray microanalysis (EDX). The particle size distribution is studied by the photon correlation spectroscopy method and the surface characteristics of prepared particles are determined by electrokinetic titration of powder suspensions using a computer-controlled system, which combines Zetamaster S of Malvern Instruments Ltd., Mettler DL 21 autotitrator and ultrasonic equipment.

3. RESULTS AND DISCUSSION

The formation of particles from vaporous phase results in obtaining homogeneous nanosized powders the characteristics of which depend on the ratio of components. The typical characteristics of prepared in similar conditions nanosized powders in the $\text{ZrO}_2(\text{Y}_2\text{O}_3)\text{-Al}_2\text{O}_3$ system are shown in Table 1.

The specific surface area of the produced powders is in the range of 29–46 m^2/g and increases with the content of alumina mainly due to the change of the specific weight of particles. Weaker dependence of the average particle size on the content of alumina is caused by the decrease of the specific weight of the particles.

The qualitative phase composition of powders is similar but quantitative phase composition depends on the content of alumina and yttria. There is a clear tendency of decrease of the m- ZrO_2 phase with the increase of Al_2O_3 content. This indicates that alumina promotes stabilization of zirconia. However, full stabilization of high temperature phases of zirconia was not achieved even at high content of alumina. The crystallite sizes of m- and t- ZrO_2 phases also decrease with the increase of Al_2O_3 (Fig. 1).

The data testify that combining synthesis of zirconia and alumina from vapour phase reduces crystallite size of both zirconia phases and content of m- ZrO_2 in prepared powders with respect to pure nanosized ZrO_2 . The crystallite size of t- ZrO_2 phase is in the range of 20–30 nm and the content of m- ZrO_2 phase is in the range of 8–16% if the powders contain more than 40 wt% of alumina. This shows that the stabilization of t- ZrO_2 phase can be achieved by the so-called size effect. Besides this, the high amount of t- ZrO_2 phase in the powders with low content of alumina and crystallite size above 30 nm indicate the dissolution of

Table 1. Characteristics of produced $\text{ZrO}_2(\text{Y}_2\text{O}_3)\text{-Al}_2\text{O}_3$ samples

No.	Chemical composition, wt%			SSA, m^2/g	<i>d</i> , nm	XRD
	ZrO_2	Al_2O_3	Y_2O_3			
1	99.8	–	–	29.3	34.8	t-, m- ZrO_2 (33%)
2	80.4	19.4	–	37.0	30.0	t-, m- ZrO_2 (20%)
3	70.5	29.4	–	39.5	29.2	t-, m- ZrO_2 (18%) θ -, δ - Al_2O_3
4	51.0	48.8	–	38.3	33.0	t-, m- ZrO_2 (16%) θ -, δ - Al_2O_3
5	24.0	75.9	–	41.4	35.0	t-, m- ZrO_2 (15%) θ -, δ - Al_2O_3
6	14.3	85.5	–	46.3	39.4	t-, m- ZrO_2 (15%) θ -, δ - Al_2O_3
7	85.1	22.2	2.5	34.9	31.8	t-, m- ZrO_2 (10%) θ -, δ - Al_2O_3
8	19.0	80.0	0.8	46.3	32.0	t-, m- ZrO_2 (8%) θ -, δ - Al_2O_3
9	–	99.8	–	44.1	37.0	θ -, δ - Al_2O_3

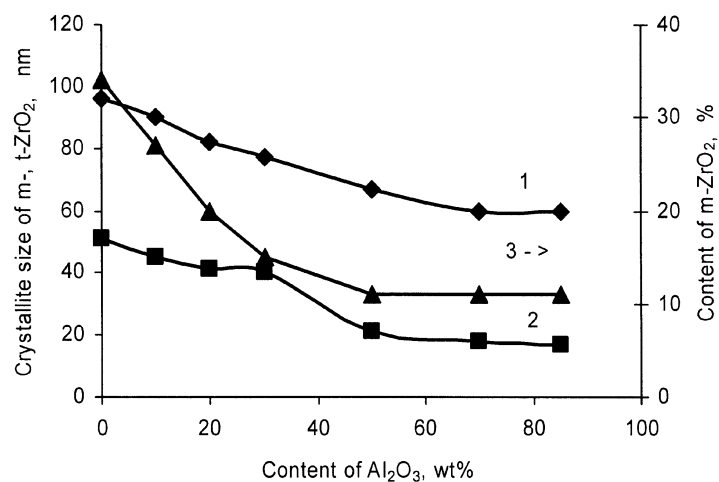


Fig. 1. Dependence of the crystallite size of m-ZrO₂ (1) and t-ZrO₂ (2) phases and of the content of m-ZrO₂ (3) in ZrO₂-Al₂O₃ powders on the content of Al₂O₃.

alumina in zirconia. ZrO₂-Al₂O₃ solid solution formation has been obtained during the hydrolysis of zirconium and aluminium izopropoxides [9].

The determined particle size distribution of ZrO₂-Al₂O₃ is practically independent of the content of alumina (Fig. 2). Contrary to the calculated data of average particle size, the particle size distribution curves show that mean particle size is around 100 nm and the size of certain particles reaches 300–800 nm.

Differences in values of the particle size can be explained by SEM studies. SEM images and X-ray microanalysis show that the produced ZrO₂-Al₂O₃ particles are highly agglomerated and form a fibrous structure (Fig. 3). However, SEM images at high magnification show that fibrous structure consists of spherical particles with size in the range of 20–60 nm (Fig. 4). Therefore different values of average particle size can be caused by the formation of agglomerated particles.

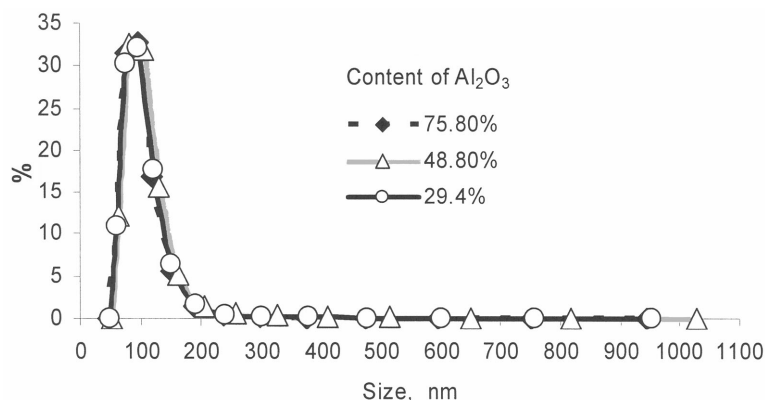


Fig. 2. Size distribution of ZrO₂-Al₂O₃ particles.

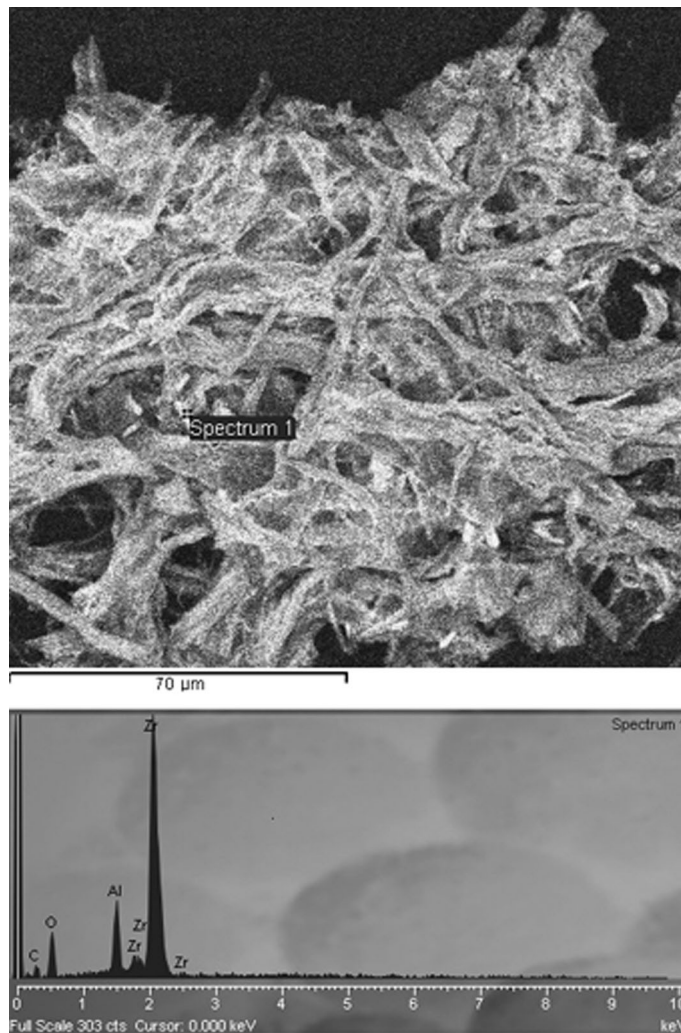


Fig. 3. SEM micrographs of $\text{ZrO}_2\text{-Al}_2\text{O}_3$ particles at low magnification and EDX spectra.

The relatively high crystallite size compared to average particle size can be explained by the presence of separate coarse particles. Such wide particle size distribution is characteristic for powders of refractory compounds, produced by thermal plasma technique, and is connected with different growth conditions of particles due to high temperature and velocity gradients of the plasma flow as well as to collisions of liquid particles or uncomplete evaporation of raw particles.

Additional calcination of $\text{ZrO}_2\text{-Al}_2\text{O}_3$ powders decreases the specific surface area and average particle size at the temperature above 900°C . Simultaneously calcination influences the crystallite size of zirconia and phase composition of powders (Fig. 5).

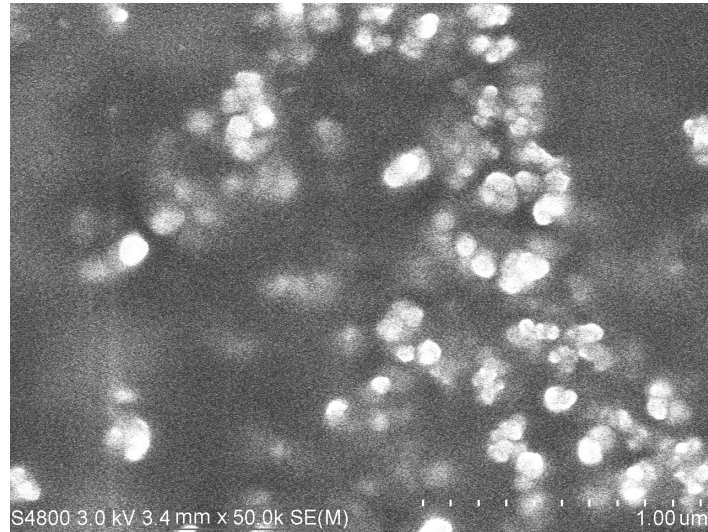


Fig. 4. SEM micrographs of $ZrO_2-Al_2O_3$ particles at high magnification.

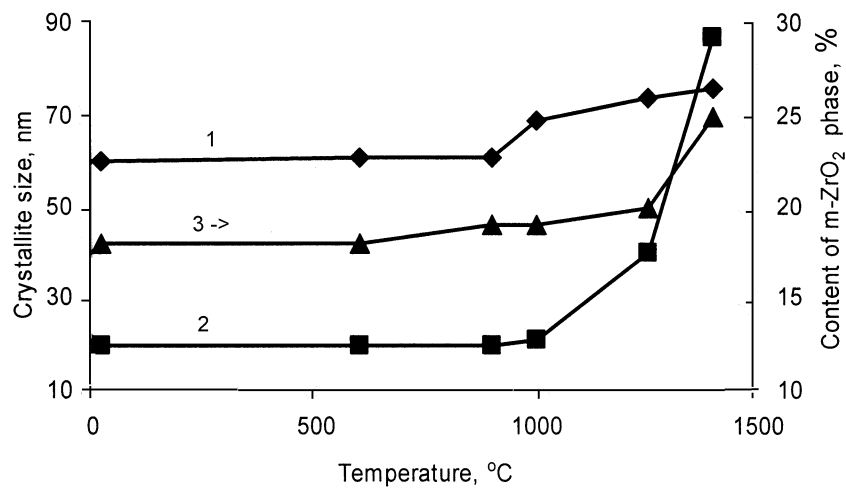


Fig. 5. Dependence of the crystallite size of m-ZrO₂ (1) and t-ZrO₂ (2) phases and content of the m-ZrO₂ (3) phase of $ZrO_2-Al_2O_3$ (78.9 wt%) on the calcination temperature.

The growth of crystallites of both zirconia phases starts at 900°C. Obviously the higher growth rate of t-ZrO₂ phase crystallites compared to m-ZrO₂ phase crystallites is connected with high content of the tetragonal phase. Despite the growing crystallite size at high temperature, the increase of the content of the m-ZrO₂ phase is insignificant. At the temperature of 1400°C the crystallite size of m-ZrO₂ and t-ZrO₂ phases is 78 nm and 85 nm, respectively, but the content of m-ZrO₂ reaches only 25%.

At similar calcination conditions, the growth of the crystallite size of pure zirconia phases starts at 500–550°C and at 1000°C the crystallite size of t-ZrO₂ and m-ZrO₂ reaches 54 nm and 86 nm, respectively. The growth of the crystallite size causes the phase transition of tetragonal modification of zirconia to monoclinic and content of the latter reaches 100% at 1200°C. Therefore the results testify that the presence of Al₂O₃ eliminates the growth of the crystallite size at the temperature up to 900°C and it stabilizes the zirconia tetragonal phase.

This feature of ZrO₂-Al₂O₃ particulate nanocomposites can be explained partially by the particle surface characteristics by electrokinetic titration (Fig. 6).

Analysis of the titration curves shows that the surface of ZrO₂ in the presence of alumina is transformed. The titration curves of ZrO₂-Al₂O₃ particles are shifted to the side of higher pH relative to pure ZrO₂ and they are close to the titration curve of Al₂O₃.

Similarity of the titration curves of ZrO₂-Al₂O₃ and Al₂O₃ samples indicates that surface characteristics of ZrO₂-Al₂O₃ particles are similar to those of alumina. This allows to suppose that the surface of zirconia is at least partially covered with alumina. Formation of such particles from the vapour phase is connected with different boiling temperatures of the compounds. The zirconia particles, formed at higher temperature, can act as nuclei for the deposition of alumina. The complex microstructure of the prepared ZrO₂-Al₂O₃ particles limits contacts between ZrO₂ particles and therefore hinders crystallite growth during calcination.

On the other hand, the high content of t-ZrO₂ phase at 1400°C and crystallite size of t-ZrO₂ phase (85 nm) indicate that stabilization of the high-temperature modification of zirconia partially can be related to the formation of the solid solution of Al₂O₃ in ZrO₂.

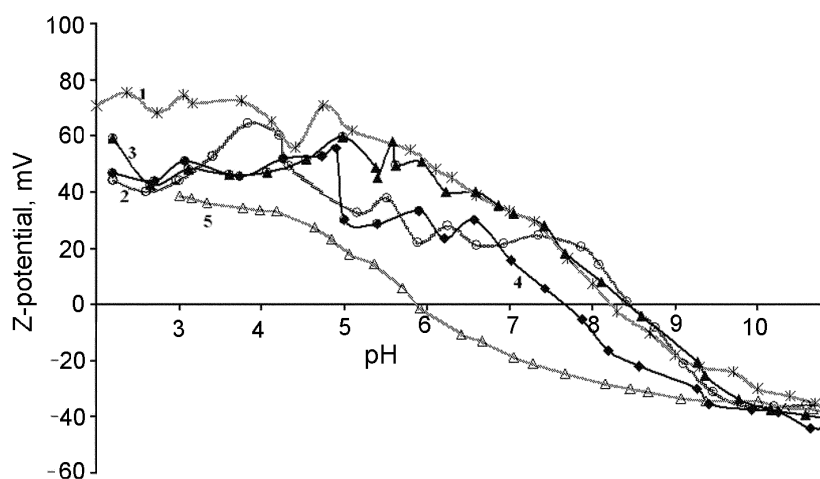


Fig. 6. Z-potential curves: 1 – ZrO₂/75.9% Al₂O₃; 2 – ZrO₂/48.8% Al₂O₃; 3 – ZrO₂/29.4% Al₂O₃; 4 – δ-, θ-Al₂O₃; 5 – t-, m-ZrO₂.

4. CONCLUSIONS

1. The developed thermal plasma technique allows to prepare spherical nano-sized $\text{ZrO}_2\text{-Al}_2\text{O}_3$ particulate nanocomposites with average particle size in the range of 30–40 nm, containing m-, t- ZrO_2 and θ -, δ - Al_2O_3 phases.
2. The presence of alumina reduces the crystallite size of zirconia phases and increases the content of the t- ZrO_2 phase relative to pure ZrO_2 , prepared in similar conditions.
3. Alumina prevents the growth of the crystallite size and phase transition t \rightarrow m- ZrO_2 during calcination at high temperature.
4. The electrokinetic titration curves testify that zirconia particles are covered with alumina.

ACKNOWLEDGEMENT

We thank Dr. A. Patmalnieks from Latvian University for SEM studies and X-ray microanalysis of the samples.

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Plasmatehnoloogia abil toodetud $ZrO_2-Al_2O_3$ osakekestest nanokomposiidid

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Jāmedateraliste oksiidide aurustamise teel plasmajoas on valmistatud nanosuuruselga (30–40 nm) sfāirilistest $ZrO_2(Y_2O_3)-Al_2O_3$ osakekestest nanokomposiidid. Āsja valmistatud pulber sisaldab m-, t- ZrO_2 ja θ -, δ - Al_2O_3 faase. On nāidatud, et Al_2O_3 olemasolu vāhendab ZrO_2 kristallide suurust ning suurendab t- ZrO_2 faasi sisaldust ja takistab ZrO_2 faaside muutumist tāiendaval kuumutamisel. $ZrO_2-Al_2O_3$ osakeste pinna uurimine kinnitab, et ZrO_2 osakeste pind on kaetud Al_2O_3 kihiga.